Synthetic cation-selective nanotube: Permeant cations chaperoned by anions

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The ability to design ion-selective, synthetic nanotubes which mimic biological ion channels may 8 have significant implications for the future treatment of bacteria, diseases, and as ultrasensitive g biosensors. We present the design of a synthetic nanotube made from carbon atoms that selectively 10 allows monovalent cations to move across and rejects all anions. The cation-selective nanotube mim-11 ics some of the salient properties of biological ion channels. Before practical nanodevices are suc-12 cessfully fabricated it is vital that proof-of-concept computational studies are performed. With this 13 in mind we use molecular and stochastic dynamics simulations to characterize the dynamics of ion 14 permeation across a single-walled (10, 10), 36 Å long, carbon nanotube terminated with carboxylic 15 acid with an effective radius of 5.08 Å. Although cations encounter a high energy barrier of 7 kT, its 16 height is drastically reduced by a chloride ion in the nanotube. The presence of a chloride ion near 17 the pore entrance thus enables a cation to enter the pore and, once in the pore, it is chaperoned by 18 the resident counterion across the narrow pore. The moment the chaperoned cation transits the pore, 19 the counterion moves back to the entrance to ferry another ion. The synthetic nanotube has a high 20 sodium conductance of 124 pS and shows linear current-voltage and current-concentration profiles. 21 The cation-anion selectivity ratio ranges from 8 to 25, depending on the ionic concentrations in the 22

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24 I. INTRODUCTION

Interest in designing ion-selective, nanoscale pores us-25 ing synthetic nanotubes to broadly mimic some of the func-26 tions of biological ion channels is rapidly gaining momentum. 27 Engineered ion channels, once successfully designed and fab-28 ricated, will have numerous potential applications, such as an-29 timicrobial agents, ultrasensitive sensors, and reverse osmo-30 sis membranes. Theoretical and experimental studies thus far 31 have mainly focused on nanotubes which can mimic the bi-32 ological water channel, aquaporin.¹⁻⁷ These synthetic water 33 channels have flow rates substantially higher than both aqua-34 porin and currently-used desalination membranes. Recently, 35 ion-selective nanotubes made from carbon atoms and boron 36 and nitrogen atoms have been successfully designed.^{8–13} By 37 placing a rim of partially charged atoms near the entrance, 38 carbon nanotubes can be made to be selectively permeable 39 to anions¹⁴ or their interior can be made hospitable specifi-40 cally to cations or anions.⁹ Similarly, Hilder et al.¹² and Won 41 and Aluru¹³ have shown ionic selectivity of boron nitride nan-42 otubes. Recently, Lee et al.15 successfully fabricated singlewall carbon nanotube ion channels 1.5 nm in diameter and 44 with negatively charged ends that are capable of selectively 45 conducting protons. By subsequent blocking of larger cations 46 they demonstrate the possibility of a single-ion detection 47 device. 48

⁴⁹ It is now possible to fabricate carbon nanotubes with ⁵⁰ either hydroxyl (-OH), carbonyl (O=C), or carboxylic acid (-COOH) functional groups attached to the end.^{1,16} Such 51 functionalized nanotubes are commercially available, have 52 high solubility,^{17,18} and are stable when embedded in a lipid 53 bilayer.¹⁹ For engineered nanotubes to have practical biologi-54 cal applications, their radii must be small and be comparable 55 to those found in biological ion channels and their length must 56 be approximately equal to the thickness of a lipid bilayer. 57 Moreover, the tubes should spontaneously insert across¹⁹ and 58 remain stable in¹⁴ the cell membrane. Once such synthetic 59 nanotubes meeting all these criteria are found, they can be 60 targeted to certain cell types by attaching specific antibodies 61 or otherwise. 62

Before practical nanoscale devices which possess the 63 functionality of biological channels can be successfully cre-64 ated and widely utilized, proof-of-concept computational 65 studies must be performed. With this aim in mind, we have 66 carried out computational studies on a functionalized car-67 bon nanotube, using molecular dynamics calculations and 68 stochastic dynamics simulations. Specifically, we examine the 69 conductance properties of a (10, 10) carbon nanotube, whose 70 effective radii and length are, respectively, 5.08 and 36 Å, 71 terminated with carboxylic acid functional groups. 72

II. COMPUTATIONAL DETAILS

We perform molecular dynamics (MD) and distributional molecular dynamics (DMD) simulations on a (10, 10) carbon nanotube terminated with carboxylic acid (-COOH) as shown in Fig. 1(a) embedded in a lipid bilayer separating two reservoirs [Fig. 1(b)]. The latter computational method combines 78

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FIG. 1. Schematic illustration of (a) the (10, 10) carbon nanotube with carboxylic acid terminated ends, and (b) the simulation assembly used in DMD, composed of reservoirs containing charged particles and the nanotube embedded in a lipid bilayer. For clarity, not all ions in the reservoir are shown. Sodium ions are shown in blue, and chloride ions in green.

⁷⁹ molecular and stochastic dynamics to reproduce the distribu ⁸⁰ tion of ion trajectories implicit in MD as closely as possible
 ⁸¹ while measuring ion permeation at much longer timescales
 ⁸² than possible with MD.²⁰ For more detail on the methods, we
 ⁸³ refer the reader to our previous papers.^{14, 20, 21}

84 A. Nanotube construction

Single-walled (10, 10) carbon nanotubes are constructed 85 from a hexagonal array of carbon atoms rolled up to form 86 a cylinder with the physical properties given by Dressel-87 haus et al.²² and a carbon-carbon bond distance of 1.42 Å.²² 88 The etching process involved in fabricating open-ended nan-89 otubes introduces either hydroxyl (-OH), carbonyl (C=O), 90 or carboxylic acid (-COOH) functional groups to the nan-91 otube ends.¹⁶ Therefore, in this paper the nanotubes are ter-92 minated with carboxylic acid (-COOH) functional groups, as 93 illustrated in Fig. 1(a). The Lennard–Jones parameters for the 94 COOH group are taken from the CHARMM27 force field²³ 95 and partial charges obtained from Zheng et al.²⁴ Specifically, 96 the partial charges for the COOH group are as follows:²⁴ C 97 (on nanotube) (0.08 e), C (0.55 e), =O (-0.50 e), O (-0.58 e), 98 and H (0.45 e) so that the nanotube (including its carboxylic 99 acid terminated ends) are overall neutral. 100

It is not clear at what pH experimental measurements 101 should be carried out to replicate the theoretical calculations 102 we describe here. As the pK_A value of the carboxylic group 103 is near 4, the terminated ends are expected to be deproto-104 nated in bulk, neutral solutions. If all carboxyl terminals are 105 deprotonated and the (10, 10) nanotube becomes terminated 106 with COO⁻, we expect the tube to be permeable predomi-107 nantly to cations. In reality, the situation is somewhat com-108 plicated in a mesoscopic system. When ionized residues are 109 located in close proximity (\sim 4–12 Å), pK values of individ-110 ual residues are known to shift away from those determined in 111 bulk solutions.^{25,26} For example, the pK_A value of four side-112 chains of the glutamic acids (E118) guarding the intracellular 113 gate of the KcsA potassium channel is calculated to be 6.9, 114 compared to the model pK_A value of 4.4.²⁵ Thus, the degree 115 of deprotonation in our COOH-terminated nanotube at vari-116 ous pH needs to be calculated by using macroscopic electro-117 statics or otherwise, and this issue deserves further theoretical 118 investigation. One of the ways to avoid this uncertainty would 119 be to replace the carboxylic functional group in the nanotube 120 with a simple ester, such as CO_2CH_3 . 121

Carbon nanotubes are defined by their chiral vector C122 $= na_1 + ma_2 = (n, m)$, where a_1 and a_2 represent the unit 123 vectors of the hexagonal lattice and n and m are integers. 124 We consider only an armchair type nanotube which is de-125 fined as (n, n). In particular, we investigate (10, 10) carbon 126 nanotubes with an effective radius of 5.08 Å, and a length of 127 approximately 36 Å. It is now technically feasible to fabri-128 cate nanotubes of this radii, and nanotubes with these func-129 tional groups (COOH) are readily available online, for ex-130 ample through CheapTubes Inc. (www.cheaptubesinc.com) or 131 NanoLab Inc. (www.nano-lab.com). Although these diameter 132 nanotubes with COOH terminated ends are available online 133 their lengths are long compared to our simulation. We require 134 lengths of 3.6 nm, whereas the shortest length available online 135 is 0.5 μ m. Therefore, to fabricate nanotubes such as those pro-136 posed here it will be necessary to shorten their length, using 137 the procedures detailed elsewhere.^{27,28} 138

B. Molecular dynamics simulations

Molecular dynamics simulations are conducted using 140 NAMD,²⁹ visualized using VMD,³⁰ with the CHARMM27 141 force field²³ and TIP3P water model. The MD domain con-142 sists of the carbon nanotube embedded in a POPE lipid bi- 143 layer separated by two reservoirs, as illustrated in Fig. 1(b). 144 The system is replicated in all three dimensions and parti-145 cle mesh Ewald electrostatics is used. The simulation box is 146 approximately $64 \times 64 \times 79$ Å², and contains 3953 water 147 molecules and either 1 sodium or 1 chloride ion. The center-148 of-mass of the nanotube is constrained to allow for tilting of 149 the nanotube, and the nanotube axis is used as the reaction 150 coordinate to measure ion trajectories. No constraints are ap-151 plied to the lipid bilayer. Each system is equilibrated for 1 ns 152 at a constant temperature of 310 K and a constant pressure of 153 1 bar. 154

The three-dimensional potential of mean force (PMF) or 155 free energy is constructed using umbrella sampling at an ionic 156

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concentration of 0 mM. In other words, only the ion of in-157 terest is present in the simulation. This enabled the three-158 dimensional PMF to be used directly in subsequent DMD 159 simulations while avoiding double counting the ion-ion in-160 teractions in subsequent DMD simulations, in which the ion-161 ion interactions are explicitly dealt with using macroscopic 162 electrostatics.¹⁴ As a result of including only the ion of in-163 terest, the system will have a non-neutral charge. With parti-164 cle mesh Ewald electrostatics this results in a homogeneous 165 neutralizing background charge filling the entire simulation 166 space.³¹ Although this will affect the free energy due to the in-167 teraction of solute charges with their own periodic copies and 168 the neutralizing background charge, its effect is negligible for 169 a non-neutral charge of ± 1 e. As mentioned the nanotube and 170 its terminated ends are overall neutral. 171

Given that the nanotube is symmetric, the ion positions 172 are sampled only in the positive z domain, and reflected to 173 obtain the negative z PMF. Thus, the ion is moved through po-174 sitions from 0 to 30 Å in steps of 0.5 Å and the z-component 175 is held using a constraint of 12.5 kcal mol⁻¹ Å⁻², while the 176 ion is free to move radially. The harmonic constraint is cho-177 sen to give adequate overlap between each window while 178 constraining the ion enough to ensure sufficient sampling. 179 The ion's z and r coordinates are obtained during each um-180 brella sampling run of 1 ns, and runs are analyzed using 181 the weighted histogram analysis method³² to obtain a two-182 dimensional PMF with the reaction coordinates (z, r). This 183 two-dimensional PMF is converted thermodynamically into a 184 three-dimensional, radially symmetric PMF as follows. Let-185 ting $U_3(z, r, \theta) = U_3(z, r)$ be the three-dimensional, radially 186 symmetric PMF and $U_2(z, r)$ be the two-dimensional cylindri-187 cal PMF, we have³³ 188

$$\exp\left(\frac{-U_2(z,r)}{kT}\right) = C_1 \int_0^{2\pi} r \exp\left(\frac{-U_3(z,r)}{kT}\right) d\theta$$
$$= 2\pi r C_1 \exp\left(\frac{-U_3(z,r)}{kT}\right), \qquad (1)$$

where C_1 is an arbitrary constant that sets the zero point of the energy. We then obtain, to within an arbitrary constant,

$$U_2(z,r) = U_3(z,r) - kT \ln(r).$$
 (2)

¹⁹¹ Thus, the three-dimensional, radially symmetric PMF can ¹⁹² be obtained from the two-dimensional cylindrical PMF ¹⁹³ by adding an offset proportional to the log of the radial ¹⁹⁴ coordinate.

195 C. Distributional molecular dynamics

The conductance of the nanotube should ideally be de-196 termined using MD which uses an explicit solvent, but it is 197 not feasible computationally to construct a current-voltage-198 concentration profile using this method. To overcome these 199 difficulties for modeling the permeation of ions across a nan-200 otube, we have devised a new methodology called DMD that 201 combines molecular and stochastic dynamics²⁰ and uses an 202 implicit solvent. The theoretical basis for this procedure and a 203 detailed test using gramicidin-A are given by Gordon et al.²⁰ 204

To begin with, molecular dynamics is used to determine 205 a free energy profile for the ion in the nanotube, and the distribution of random and frictional forces (friction kernel) is 207 measured over discrete segments of the tube. By doing this 208 we can match the distribution of ion trajectories in DMD as 209 closely as possible to MD. These parameters are then used in 210 stochastic dynamics simulations based on the nonlinear generalized Langevin equation given by 212

$$\partial_t \mathbf{q}(t) = m^{-1} \mathbf{p}(t), \tag{3}$$
$$\partial_t \mathbf{p}(t) = \mathbf{F}_{\mathbf{D}}(\mathbf{q}(t)) - \int_0^t dt' K(t') \mathbf{p}(t-t') + \mathbf{F}_{\mathbf{R}}(t), \tag{3}$$

$$\left\langle \mathbf{F}_{\mathbf{R}}(0)\mathbf{F}_{\mathbf{R}}^{T}(t)\right\rangle = kTK(t)m.$$
(4)

The nonlinear generalized Langevin equation is solved using 220 an extension of the leapfrog method as outlined in Gordon 221 $et al.^{34}$ (further detail is also provided by Nilsson and Padró³⁵ 222 and Wan *et al.*³⁶). 223

In order to prevent double-counting of the interaction between any given ion and the net effect of all the other ions, we perform our molecular dynamics simulation at 0 mM to generate the PMF used in DMD. Preliminary work suggests that the PMF obtained from DMD at finite concentrations compares well with that obtained from MD. All ion-ion related effects are a result of the macroscopic electrostatics that are built into the stochastic simulation.

The simulation space is divided into (i) a channel region, 232 where the nonlinear generalized Langevin Eq. (3) is solved 233 with a 2 fs timestep using the free energy and friction ker-234 nel determined from MD; and (ii) the bulk region, where nor-235 mal Brownian dynamics and macroscopic electrostatics are 236 performed with a longer timestep of 100 fs. All single-ion 237 deterministic forces are derived from the PMF taken from 238 MD, which represent the interactions between a single ion 239 and the nanotube/membrane system. In contrast, the ion-ion 240 interactions are calculated using macroscopic electrostatics 241 (Poisson's equation with a dielectric barrier defined by the 242 nanotube/membrane system) and other fitted short-range ion- 243 ion forces. This approach may ignore the effect of anisotropic 244 permittivity of water within the channel demonstrated by 245 Toghraee *et al.*³⁷ However, in our simulations the anisotropic 246 behavior only applies to the ion-ion interaction as all single 247 ion forces are derived from our preceding MD simulations. 248 Therefore, it is expected that the effect of anisotropy on our 249 results will be marginal. In most cases the ions reside on the 250 central axis of the pore where the axial permittivity is equiv-251 alent to bulk water,³⁷ and therefore axial ion-ion interactions 252 will be unaffected by the anisotropic behavior. We note that 253 the value used in our simulations for the dielectric constant 254 of water has been shown to compare well to the physiological 255 conductance through biological potassium channels.³⁸ 256 In MD, the COOH ends are free to move. In DMD a rigid
tube boundary generated from equilibrated MD is present, but
this boundary is only applied as a dielectric boundary for the
calculation of ion-ion interactions. In contrast, the interactions
between single ions and the tube walls are determined by the
PMF, which represents the time-averaged effect of the flexible
tube on the ions.

As mentioned, the PMF was determined using WHAM.³² The data gathered for the WHAM is also used to determine the friction kernel, K(t). For each ion trajectory we determine the momentum autocorrelation function C(t) and derive K(t)using

$$\partial_t C = -\int_0^t K'(t - t')C(t') \, dt'.$$
 (5)

The inverse velocity decay time due to friction γ is then determined as the time integral of K(t) from t = 0 to $t = \infty$. Note that the diffusion coefficient, *D* can be determined using γ and the Einstein relation $D = kT/(m\gamma)$, where *k* is Boltzmann's constant, *T* is the temperature, and *m* is the mass. We then assume the friction kernel can be approximated by an exponential function, thus

$$K(t) = \gamma \kappa \exp(-\kappa t), \tag{6}$$

²⁷⁶ so as to determine the inverse decay time of the friction mem-²⁷⁷ ory kernel, κ .

In summary, three parameters that feature in the gener-278 alized Langevin Eq. (3) are the deterministic force $\mathbf{F}_{\mathbf{D}}$, the 279 random force F_R , and the frictional force F_F . The latter two 280 forces represent the stochastic force arising from random col-281 lisions of an ion with its surrounding molecules. In Brownian 282 dynamics these parameters are estimated by solving Poisson's 283 equation and by using the fluctuation-dissipation theorem. In 284 DMD, the pore is divided into thin slices, and the forces acting 285 on an ion and the diffusion coefficient in each section are ob-286 tained from a three-dimensional PMF and friction kernel de-287 termined from MD. This ensures that for a single ion certain 288 properties of the trajectory in MD, such as the probability of 289 finding an ion at a given position, will be reproduced in DMD. 290 When there are two ions in the pore, the interactions between 291 them are treated using macroscopic electrostatics. Poisson's 292 equation is solved to determine the ion-ion interaction with 293 the results for all possible configurations of the two ions be-294 ing stored in a lookup table which is consulted throughout the 295 simulation.²¹ 296

All current-voltage curves were generated at an ionic 297 concentration of 500 mM, and all current-concentration pro-298 files were generated at a voltage of 200 mV. Voltage is ap-299 plied across the membrane in DMD simulations. A uniform 300 electric field is applied across the channel, equivalent to plac-301 ing two voltage plates at a distance and applying a potential 302 $(\pm V)$ on each plate. Current was calculated using the formula 303 $= qn/\Delta t$, where n is the average number of ions that cross I 304 the membrane, q is the charge of the ion, and Δt is the simula-305 tion time of one run, which in our case is 0.8 μ s. The numbers 306 of ions crossing the membrane are counted in the simulation 307 period of 0.8 μ s, and the simulation was repeated 5 times. 308 The average number of ions that cross the membrane and the 309 standard error of the mean are determined from these five sim-310

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ulations. The starting configuration of ions in the simulation is 311 generated randomly, and each simulation is given a different 312 randomly generated distribution. No ions are present inside 313 the channel at the beginning of the simulation. 314

III. RESULTS AND DISCUSSION

The free energy profiles encountered by sodium (Na⁺) 316 and chloride (Cl⁻) as they traverse the (10, 10) carbon nan-317 otube with its ends terminated with carboxylic acid sug-318 gests that the pore will be impermeable to charged particles. 319 Figure 2 shows the free energy profiles at 0 mM ionic con-320 centration of sodium and chloride ions as they traverse the 321 (10, 10) carbon nanotube with carboxylic acid terminated 322 ends. Displayed in the upper panel [Figs. 2(a) and 2(b) are 323 the one-dimensional free energy profiles, obtained from the 324 two-dimensional (z, r) PMF by thermodynamic integration, 325 of sodium and chloride ions, respectively. The lower panel 326 [Figs. 2(c) and 2(d)] shows the two-dimensional free energy 327 contours generated from the three-dimensional PMF by tak-328 ing a slice through the nanotube center. The upper panels of 329 Fig. 2 are one-dimensional PMFs and do not simply represent 330 the on-axis values from the lower panels of Fig. 2. Instead, 331 they are integrations over all radial and angular values avail-332 able to the ion into a single value for each z coordinate of the 333 ion.³³ If $U_3(x, y, z)$ is the three-dimensional PMF and $U_1(z)$ is 334 the one-dimensional PMF, then 335

$$U_1(z) = -kT \ln\left(L_0^{-2} \int dx \, dy [\exp(-U_3(x, y, z)/kT)]\right),$$
(7)

where L_0 is an arbitrary length scale which simply determines ³³⁶ the zero point of the energy. ³³⁷

For sodium ions, there exists a free energy barrier of approximately 7 kT, as illustrated in Figs. 2(a) and 2(c). At the entrance and exit of the nanotube (at $z = \pm 16$ Å) there exist two attractive energy wells for sodium ions. From Fig. 2(a) this well appears as a slight dip, but for Fig. 2(c) the well is more distinct (purple region) and is approximately 6 kT.

Similarly, a chloride ion encounters a free energy barrier 344 of approximately 7 kT, as illustrated in Fig. 2(b). However, 345 unlike sodium, inside the nanotube chloride ions encounter a 346 reduction in free energy well of approximately 4.5 kT, as illus-347 trated in Fig. 2(d) (blue). Therefore, once inside the nanotube 348 chloride ions encounter an energy barrier to exit of approxi-349 mately 4.5 kT [Fig. 2(b)]. Just outside the nanotube entrance 350 and exit (at $z = \pm 19$ Å) chloride ions encounter an energy 351 well of 2 kT, as is evident in Fig. 2(b). From Fig. 2(d) this 352 energy well is shown to occur approximately 4 Å from the 353 nanotube central axis (dark blue). Just inside the nanotube en-354 trance and exit (at $z = \pm 14$ Å) a chloride ion encounters an 355 energy well of approximately 2 kT, which is shown as a slight 356 dip in Fig. 2(b), and a distinct well in Fig. 2(d) (blue). How-357 ever, in contrast to sodium the energy at the nanotube entrance 358 increases rapidly as the chloride ion moves away from the ax-359 ial center of the nanotube. As a result a chloride ion encoun-360 ters a much narrower entrance to the nanotube than a sodium 361 ion. The terminated ends of the nanotube (COOH groups) are 362



FIG. 2. Free energy of sodium (Na^+) and chloride (Cl^-) ions through the (10, 10) carbon nanotube with carboxylic acid terminated ends at 0 mM ionic concentration. Upper panel: one-dimensional free energy profile of (a) sodium and (b) chloride. Energy in the upper panel is given in units of kT (left-hand side) and kcal/mol (right-hand side). Lower panel: two-dimensional contour plot of the free energy landscape (kT) within the nanotube for (c) sodium and (d) chloride.

³⁶³ overall neutral and represent tight dipoles at the entrance and ³⁶⁴ exit of the nanotube. Therefore, as expected the electric field ³⁶⁵ decays quickly, as $1/R^3$.

Although the free energy barriers encountered by both 366 the sodium and chloride ions are generally insurmountable, 367 our simulations reveal that in fact substantial numbers of 368 sodium ions move across the channel. As shown in Fig. 3(a), 369 for sodium ions the current is 12 pA at an applied potential 370 of 100 mV and ionic concentration of 500 mM. In contrast, 371 for chloride ions the current is negligible at only 0.3 pA at 372 100 mV. By fitting a linear regression to the data points we 373 obtain a large conductance of 124 ± 4 pS for sodium ions 374 and 5 ± 0.6 pS for chloride ions. Alternatively, with an ionic 375 concentration of 200 mM, the sodium and chloride current at 376 100 mV is 5.3 and 0.64 pA, respectively. On fitting a linear 377 regression to the data points we obtain a conductance of 55 \pm 378 2 pS for sodium ions and 7 ± 0.4 pS for chloride ions (data 379 not shown). The chloride conductance remains a similar mag-380 nitude despite the reduction in ionic concentration from 500 381 to 200 mM. The energy well encountered by sodium ions at 382 the nanotube entrance [Figs. 2(a) and 2(c)] acts to limit the 383 number of chloride ions which can pass through the nanotube 384 since for chloride ions to pass through the nanotube entrance 385 they must follow the axial center of the nanotube. 386

Figure 3(b) shows the current-concentration profile of 387 sodium ions. The current is shown to steadily increase with 388 increasing ionic concentration following a linear relationship 389 rather than that of Michaelis-Menten form. This suggests that 390 the time spent by an ion waiting to enter the channel will de-391 crease as the concentration increases. Even at high concen-392 trations, the rate limiting step is the diffusion of ions into the 393 channel. 394

To clarify how sodium ions can conduct across this nan-395 otube despite an insurmountable energy barrier observed in 396 Fig. 2 we examine the ion binding sites within the nanotube, 397 how a resident ion transits from one binding site to another 398 and study ion animations of the conduction process. We find 399 that cations are able to move across the (10, 10) carbon nan-400 otube pore only when they are chaperoned by chloride ions. 401 At the nanotube entrance a resident sodium ion enables chlo-402 ride to overcome the energy barrier and enter the nanotube 403 interior. Once inside the nanotube these resident chloride ions 404 act to ferry, or chaperone, sodium ions from the entrance to 405 the exit of the nanotube. This artificial channel resembles that 406 of the mutant glycine receptor³⁸ in which anions act to chaperone sodium ions^{40,41} with a conductance of 17 pS. 408

We divide the nanotube length into 100 thin sections and 409 compute time averages of both sodium and chloride ions in 410 each section during the simulation period of 4 μ s. In the 411 absence of an applied potential, resident chloride ions tend 412 to dwell within the nanotube and resident sodium ions tend 413 to dwell at the pore entrances and within the nanotube, as 414 shown in Fig. 4. On average, there are 1.7 chloride ions oc-415 cupying the nanotube interior. In contrast, there are on aver-416 age 0.7 sodium ions occupying the binding sites at the pore 417 entrances, and 1.1 sodium ions occupying the channel inte-418 rior. Although almost two chloride ions occupy the channel 419 during the simulation the chloride conductance is consider-420 ably less than sodium, 5 pS compared to 124 pS, suggesting 421 that chloride is acting to mainly chaperone sodium across the 422 channel. 423

Detailed analysis of the dynamics of ion permeations reveals that translocation of sodium ions across the nanotube is indeed achieved by ion chaperoning. This process can be



FIG. 3. Current-voltage-concentration profiles for the (10, 10) carbon nanotube with carboxylic acid terminated ends. (a) Current-voltage profile for sodium ions using an ionic concentration of both 200 (open circles) and 500 mM (filled circles), and chloride ions using an ionic concentration of 500 mM (filled triangles). (b) Current-concentration profile for sodium at an applied potential of 200 mV. Error bars represent one standard error of the mean, and error bars smaller than the data points are not shown.

observed in DMD animations (snapshot is shown in Fig. 5) in 427 which a resident chloride ion acts to ferry sodium ions from 428 the entrance to the exit of the nanotube. We can also detail 429 this process by recording the axial position of every ion in the 430 pore at each timestep under the influence of an applied po-431 tential of 100 mV and at an ionic concentration of 500 mM. 432 These positions can then be used to determine the ion con-433 figuration within the nanotube. Initially sodium ions occupy 434 the two binding sites at the nanotube entrance and exit. These 435 sodium ions enable a chloride ion to overcome its energy bar-436 rier and enter the nanotube interior. However, this event takes 437 on average 50 ns. A chloride ion then occupies the nanotube 438 interior for the remainder of the simulation. Sodium ions can 439 only enter the nanotube interior when the resident chloride is 440 present. Therefore, for the first 50 ns of simulation no sodium 441 ions are present in the nanotube interior. Sodium ions then 442 occupy at least one binding site for the remainder of the sim-443 ulation. In fact, for 72% of the simulation time a sodium ion 444



FIG. 4. Binding sites for (a) sodium and (b) chloride ions for the (10, 10) carbon nanotube with carboxylic acid terminated ends in the absence of an applied potential and at an ionic concentration of 500 mM.

occupies an entrance and/or exit binding site and the nanotube 445 interior. 446

The presence of a chloride ion in the pore lowers the 447 energy barrier encountered by the sodium ion. Figure 6 il- 448 lustrates how a resident chloride ion, fixed at the nanotube 449



FIG. 5. A segment of video animation from DMD simulations showing a resident chloride ion (gold) which acts to chaperone sodium ions (pink) from entrance (left) to exit (right) of the pore (enhanced online). For clarity the nanotube is shown as a bronze surface with half of the channel removed.



FIG. 6. One-dimensional free energy profile of sodium ions through the (10, 10) carbon nanotube with carboxylic acid terminated ends with (i) no chloride ions (Cl⁻) present and (ii) a chloride ion fixed at z = 0 Å.

center or z = 0 Å, reduces the energy barrier encountered by 450 an approaching sodium ion. When no chloride ion is present 451 a sodium ion encounters an energy barrier of approximately 452 7 kT. However, the sodium ion no longer experiences an en-453 ergy barrier when a chloride ion is present in the nanotube 454 interior. Instead, the energy 455 exit increases from 2 to 4.5 greathese wells are separated by 456 a central barrier of approximately 4 kT. A free energy barrier 457 of 2 kT remains for a sodium ion to enter the entrance or exit 458 binding sites. 459

IV. CONCLUSION 460

In summary, we show that a (10, 10) carbon nanotube 461 with an effective radius of 5.08 Å and a length of 36 Å termi-462 nated with carboxylic acid (COOH) are selective to sodium 463 ions despite both sodium and chloride encountering an en-464 ergy barrier of 7 kT to enter the nanotube. We find that chlo-465 ride ions act as chaperones for sodium ions. The nanotube 466 has a sodium conductance of 124 pS, and a negligible chlo-467 ride conductance of 5 pS. Once inside the nanotube interior, 468 chloride ions ferry sodium ions from the entrance to the exit. 469 This artificial channel resembles that of the mutant glycine 470 receptor³⁹ in which anions act to chaperone sodium ions with 471 a conductance of 17 pS. This work illustrates the danger of re-472 lying solely on the single particle free energy profile to form 473 physical insight into the ion conduction mechanisms. а 474

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A. Noy, and O. Bakajin, Proc. Natl. Acad. Sci. U. S. A. 105, 17250 (2008). 485

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547

- ummer, J. C. Rasaiah, and J. P. Noworyta, Nature 414, 188 (2001). 486 A. Waghe, J. C. Rasaiah, and G. J. Hummer, Chem. Phys. 117, 10789 487 (2002)488
- ⁴A. Kalra, S. Garde, and G. Hummer, Proc. Natl. Acad. Sci. U. S. A. 100, 489 10175 (2003). 490
- ⁵B. Corry, J. Phys. Chem. B **112**, 1427 (2008).
- ⁶J. A. Garate, N. J. English, and J. M. D. MacElroy, Mol. Simulat. 35, 2 492 (2009)493
- ⁷T. A. Hilder, D. Gordon, and S. H. Chung, Small 5, 2183 (2009).
- ⁸C. Peter and G. Hummer, Biophys. J. 89, 2222 (2005).
- ⁹S. Joseph, R. J. Mashl, E. Jakobsson, and N. R. Aluru, Nano Lett. 3, 1399 496 (2003)497
- ¹⁰M. Majumder, N. Chopra, and B. J. Hinds, J. Am. Chem. Soc. 127, 9062 498 (2005).499
- ¹¹J. H. Park, S. B. Sinnott, and N. R. Aluru, Nanotech. 17, 895 (2006).
- ¹²T. A. Hilder, D. Gordon, and S. H. Chung, Small 5, 2870 (2009).
- ¹³C. Y. Won and N. R. Aluru, Chem. Phys. Lett. **478**, 185 (2009).
- ¹⁴T. A. Hilder, D. Gordo S. H. Chung, Biophys. J. **99**, 1734 (2010). ¹⁵C. Y. Lee, W. Choi, J. **7**T. Han, and M. S. Strano, Science **329**, 1320 503 504 (2010).505
- ¹⁶T. Lin, V. Bajpai, T. Ji, and L. Da t. J. Chem. **56**, 635 (2003).
- ¹⁷J. L. Bahr and J. M. Tour, J. Mater Chem. **12**, 1952 (2002).
- ¹⁸D. Tasis, N. Tagmatarchis, V. Georgakilas, and M. Prato, Chem. Eur. J. 9, 508 4000 (2003) 509
- ¹⁹C. F. Lopez, S. O. Nielsen, P. B. Moore, and M. L. Klein, Proc. Natl. Acad. 510 Sci. U. S. A. 101, 4431 (2004). 511
- ²⁰D. Gordon, V. Krishnamurthy, and S. H. Chung, J. Chem. Phys. 131, 512 134102 (2009). 513
- ²¹M. Hoyles, S. Kuyucak, and S. H. Chung, Phys. Rev. E 58, 3654 (1998). 514 515
- ²²M. S. Dresselhaus, G. Dresselhaus, and R. Saito, Carbon 33, 883 (1995).
- ²³B. R. Brooks, R. E. Bruccoleri, B. D. Olafson, D. J. States, S. Swaminathan, 516 and M. J. Karplus, Comput. Chem. 4, 187 (1983); A. D. MacKerell, Jr, B. 517 Brooks, C. L. Brooks III, L. Nilsson, B. Roux, Y. Won, and M. Karplus, 518 in The Encyclopedia of Computational Chemistry, edited by PvR Schleyer 519 et al., Vol. 1, pp. 271-277 (John Wiley & Sons, Chichester), (1998). 520
- ²⁴J. Zheng, E. M. Lennon, H. K. Tsao, and S. Jiang, J. Chem. Phys. 122, 521 214702 (2005). 522
- ²⁵K. M. Ranatunga, I. H. Shrivastava, G. R. Smith, and M. S. P. Sansom, 523 Biophys. J. 80, 1210 (2001). 524
- ²⁶C. Adcock, G. R. Smith, and M. S. P. Sansom, Biophys. J. 75, 1211 525 (1998)526
- ²⁷K. J. Ziegler, Z. Gu, J. Shaver, Z. Chen, E. L. Flor, D. J. Schmidt, C. Chan, 527 R. H. Hauge, and R. E. Smalley, Nanotechnology 16, S539 (2005) 528
- ²⁸S. Wang, Z. Liang, B. Wang, C. Zhang, and Z. Rahman, Nanotechnology 529 18,055301 (2007). 530
- ²⁹L. Kalé, R. Skeel, M. Bhandarkar, R. Brunner, A. Gursoy, N. Krawetz, J. 531 Phillips, A. Shinozaki, K. Varadarajan, and K. Schulten, J. Comput. Phys. 532 151, 283 (1999). 533
- ³⁰W. Humphrey, A. Dalke, and K. Schulten, J. Molec. Graphics 14, 33 534 (1996). 535
- ³¹M. A. Kastenholz and P. H. Hünenberger, J. Phys. Chem. B 108, 774 536 (2004)537
- ³²S. Kumar, J. M. Rosenberg, D. Bouzida, R. H. Swendsen, and 538 P. A. Kollman, J. Comput. Chem. 16, 1339 (1995); A. Grossfield, 539 http://membrane.urmc.rochester.edu/Software/ WHAM/WHAM.html 540 [accessed 20 August 2008]. 541
- ³³H. J. C. Berendsen, Simulating the Physical World: Hierarchical Modeling 542 from Quantum Mechanics to Fluid Dynamics (Cambridge University Press, 543 Cambridge, 2007). 544
- ³⁴D. Gordon, V. Krishnamurthy, and S. H. Chung, Molecular Physics 106, 545 1353 (2008). 546
- ³⁵L. G. Nilsson and J. A. Padró, Molecular Physics 71, 355 (1990).
- ³⁶S. Z. Wan, C. X. Wang, and Y. Y. Shi, Molecular Physics 93, 901 548 (1998)549
- ³⁷R. Toghraee, R. J. Mashl, K. I. Lee, E. Jakobsson, and LL Ravaioli, J. 550 Comput. Electron. 8, 98 (2009). 551
- ³⁸J. A. Ng, T. Vora, V. Krishnamurthy, and S. H. Chung, Biophys. J. 37, 213 552 (2008).553
- ³⁹M. O'Mara, P. H. Barry, and S. H. Chung, Proc. Natl. Acad. Sci. U. S. A. 554 100, 4310 (2003). 555
- ⁴⁰S. Sugiharto, T. M. Lewis, A. J. Moorhouse, P. R. Schofield, and P. H. 556 Barry, Biophys. J. 95, 4698 (2008). 557
- ⁴¹P. H. Barry, S. Sugiharto, T. M. Lewis, and A. J. Moorhouse, Channels 4, 558 142 (2010). 559